

1 **Upcycling spent coffee grounds and waste PET bottles into electrospun**
2 **composite nanofiber mats for oil structuring applications**

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22 **Abstract**

23 Spent coffee grounds (SCG) and post-consumer PET bottles (PETbot) are major waste
24 materials from the food industry. In this work, their use to obtain electrospun composite
25 nanofibers is explored aiming to develop a new route for their upcycling into high
26 added-value oil-based products. Raising the proportion of PETbot increased the average
27 fiber diameter (from 0.36 to 0.89 μm) and enhanced the tensile properties of the mats.
28 Young's modulus and strain-at-break increased from 5.1 to 32.8 MPa and from 13.8 to
29 39.4%, respectively, as the SCG:PETbot weight ratio decreased. Dispersions of the mats
30 in castor oil exhibited shear-thinning and gel-like viscoelastic properties which can be
31 tuned through the SCG:PETbot ratio. Friction coefficient values of resulting oleogels
32 tested in a tribological contact ranged from 0.165 to 0.092. SCG and PETbot can be
33 used to obtain electrospun nanofibers with oil structuring capabilities while resulting
34 oleogels are proposed as environmentally-friendly alternatives to semisolid lubricants.

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36 **Keywords:** Electrospinning, Lignocellulose, Nanofibers, Oleogels, PET, Rheology,
37 Waste.

38 **1. Introduction**

39 There is increasingly widely awareness that the traditional linear economy model, by
40 which resources are collected and transformed into products and consumer goods for
41 use until they are disposed of as waste, is making our planet unsustainable (Michelini et
42 al., 2017). In stark contrast to nature's life cycle, the still prevailing linear economic
43 system focuses on the short-term consumption, thus precluding sustainable development
44 (Sarja et al., 2021). The concept of circular economy, inspired by nature's cyclical
45 model, emerged strongly in the past decade to solve the ensuing problems. In circular
46 economy, resources are managed and used more sustainably by having products and
47 services in circulation for as long possible. The primary aim is to minimize production
48 and, when a given product must unavoidably be used, to recycle and reuse it. Circular
49 economy also advocates using as many biodegradable and/or renewable materials as
50 possible (Camacho-Otero et al., 2018; Geisendorf and Pietrulla, 2018) to produce
51 consumer goods. In those cases where using environmentally friendly materials or
52 components is impossible, the aim is to give existing products a second life by recycling
53 or, preferably, upcycling, i.e. by reinserting them into the production cycle in order to
54 develop new added-value products (Morseletto, 2020; Webster, 2021).

55 The food, chemical and textile industries generate vast amounts of waste from
56 the start of the production cycle through packaging to disposal by consumers (Reynolds
57 et al., 2016; Sridhar et al., 2021; Thyberg and Tonjes, 2016). One salient example of
58 waste from the food industry is spent coffee grounds (SCG) (Campos-Vega et al.,
59 2015). This lignocellulosic waste has become an environmental concern because, unless
60 disposed of appropriately, it can release caffeine, tannins and polyphenols resulting
61 from coffee fermentation (Bomfim et al., 2022; Franca and Oliveira, 2022). SCG are
62 generated as waste after infusion and account for about 90% of the initial mass of the
63 beans. In 2022, global coffee production exceeded 10 million tons (Bomfim et al., 2022;
64 Forcina et al., 2023). A number of efforts have been made over the last decade to
65 valorize and/or repurpose this waste in eco-friendly ways in fields such as agriculture
66 and biofuel energy production, and also in the production of polymer and composite

67 precursors (Kang et al., 2023; Nanni et al., 2022; Gupta et al., 2022), graphene
68 (Dericiler et al., 2022) and other chemicals (Leow et al., 2021), dye sorbents (Kasbaji et
69 al., 2023) or 3D printing materials (Yu et al., 2023), among other uses (Jin Ong et al.,
70 2023; Pagett et al., 2023).

71 Plastic is one of the most common food packaging wastes from the food industry
72 (Evode et al., 2021; Ncube et al., 2021). In fact, its role in food safety and preservation
73 has turned it into an indispensable convenience. Global plastic production in 2021
74 amounted to 390.7 million metric tons, 4% over the figure for the previous year
75 (Statista, 2023). Also, it is expected to continue to rise sharply and reach around 1.1
76 billion tons by 2050 (Naderi Kalali et al., 2023). The US Environmental Protection
77 Agency (EPA) reports that only 7% of all plastic is recycled, of which 30% corresponds
78 to used poly(ethylene terephthalate) (PET) bottles (Naderi Kalali et al., 2023).

79 Efficient recycling of plastic polymer waste has become a major priority
80 worldwide (Shanmugam et al., 2020). In fact, post-consumer PET bottles (PETbot) are
81 highly valued for producing a variety of polymeric and composite materials because
82 they are much less expensive than virgin PET (Rosenboom et al., 2022; Smith et al.,
83 2022). Recycled PET has thus emerged as a highly suitable, cost-effective choice for a
84 wide variety of uses such as protective clothing, membranes, vascular grafts, tissue
85 scaffolding and filtration media (Rahmati Nejad et al., 2020; Sadeghi et al., 2021; Tian
86 and Mo, 2019).

87 Oleogelation and oil structuring have aroused much interest for a broad range of
88 uses (Davidovich-Pinhas, 2016; Pakseresht and Mazaheri Tehrani, 2022; Patel, 2018,
89 2017) but particularly in the production of lubricants, coatings and adhesives (Gallego
90 et al., 2015a; Raghunanan et al., 2018; Borrero-López et al., 2018; Tenorio-Alfonso et
91 al., 2021, 2020). Increasing environmental awareness has raised a growing industrial
92 concern with replacing petroleum based components with renewable and/or recycled
93 materials. For instance, the lubricant industry is not only replacing conventional mineral
94 oils with vegetable oils or glycerol esters (Panchal et al., 2017; Syahir et al., 2017), but
95 also pursuing the development of semisolid lubricants and greases obtained by using

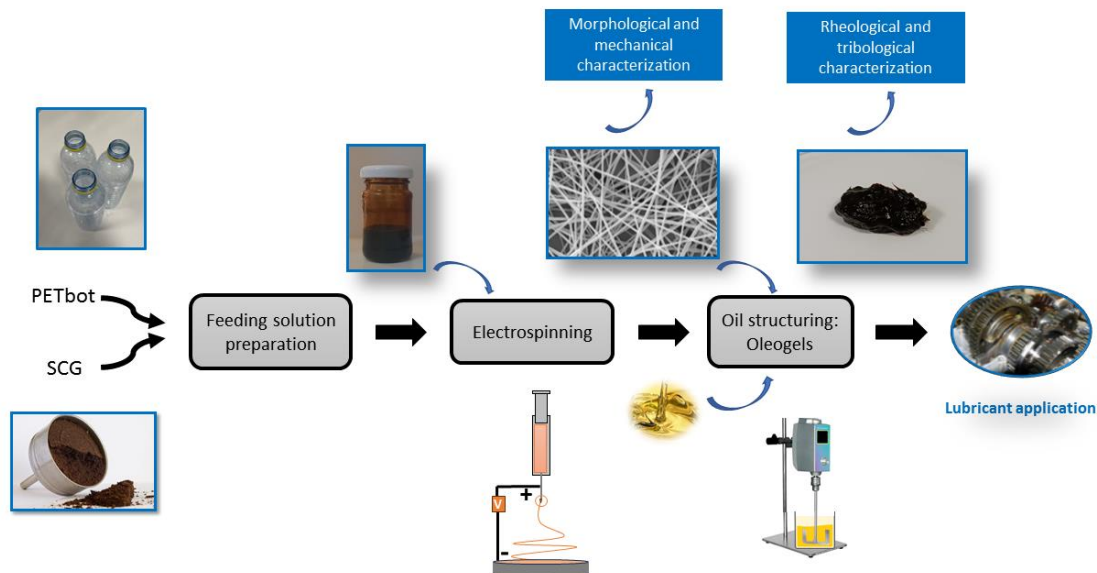
96 environmentally friendly oil structuring and thickening agents with specific rheological
97 and tribological functionalities. This functional performance has traditionally been
98 achieved by using metal soaps —lithium soaps, mainly (~80%)— and polyureas or
99 other polymers from the petrochemical industry. However, given that global grease
100 production is estimated at around 1.1 Mton/year (Lubes'n'Greases, 2021), and that the
101 thickener content in greases typically ranges from 3% to 40% by weight, the search for
102 technically efficient renewable or bio-sourced thickening agents is a challenging goal.
103 One must also consider the significance of lithium to other more demanding and
104 pressing uses such as batteries and the supply constraints on this scarce resource.
105 Similar considerations also apply to oil-based adhesive and coating formulations
106 (Bashir et al., 2022; Cheng et al., 2018; Sahoo et al., 2018; Tenorio-Alfonso et al.,
107 2020). New oil structuring agents based on renewable and/or recycled materials are
108 therefore needed to meet existing technical and environmental challenges.

109 In previous work, different approaches involving the chemical alteration of
110 biopolymers to induce oil structuring via chemical crosslinking with isocyanate or
111 epoxide reactive groups were followed (Borrero-López et al., 2017; Cortés-Triviño et
112 al., 2021, 2018; Gallego et al., 2015a, 2013). Nevertheless, these modifications usually
113 require using solvents and chemical reagents that make the synthesis of oil thickeners
114 somewhat cumbersome and not entirely environmentally friendly. Very recently,
115 nanostructures obtained by electrospinning unmodified lignocellulosic polymers to be
116 effective oil structuring agents were assessed (Borrego et al., 2022, 2021; Rubio-Valle
117 et al., 2023); however, some cospinning polymer was occasionally required to improve
118 the morphology of the resulting electrospun nanofiber mats. The aim of this work is to
119 investigate the use of a lignocellulosic waste material such as spent coffee grounds
120 (SCG) to obtain electrospun nanofiber mats with potential oil structuring capabilities.
121 Also post-consumer PET bottles were evaluated as a cospinning polymer to develop a
122 new valorization or upcycling route for these waste materials that is fully aligned with
123 the circular economy paradigm. It is the hypothesis of this work that the high aspect
124 ratio and porosity of SCG/PETbot nanofiber webs can promote the formation of 3D

125 percolation networks that enhance the physical interactions with the oil and help
 126 stabilize the colloidal system. Moreover, the rheological properties of the oleogels
 127 derived should be tunable by varying the SCG:PETbot proportion. Finally, a potential
 128 application of these oleogels as semi-solid lubricants is explored in this work. Although
 129 PET from bottles has already been used as a component of electrospun membranes for
 130 various uses (Hossain et al., 2022; Strain et al., 2015), SCG has to the best of our
 131 knowledge never been used before to obtain electrospun nanofiber mats. In any case,
 132 neither material has been explored as an oil structuring agent for producing gel-like or
 133 soft materials. For a better and quick understanding of the whole process aimed at
 134 upcycling SCG and PETbot waste materials into oil structuring agents, finally leading to
 135 oleogels with potential lubricant applications, Scheme 1 displays a flow chart in which
 136 the sequence of the different stages followed can be readily visualised.

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139 **Scheme 1.** Synopsis of the process to upcycle SCG and PETbot into oil
 140 structuring agents for lubricant applications and characterization methodologies
 141 followed.

142

143 **2. Material and methods**

144 *2.1. Materials*

145 SCG was collected from a local cafe in Huelva (Spain). The material was obtained by
146 brewing Spanish Delta coffee and oven-drying it at 100 °C for 24 h. Previously
147 collected post-consumer beverage bottles (PETbot, Fuenteliviana, Spain) were cleaned
148 with household detergent and ultrapure water, dried in an oven at 100 °C for 24 h and
149 cut into small (1–2 mm) pieces. Trifluoroacetic acid (TFA) for synthesis and
150 dichloromethane (DCM, purity $\geq 99.5\%$), both supplied by Merck Sigma–Aldrich
151 (Darmstadt, Germany), were used as solvents. Castor oil was purchased from Guinama
152 (Valencia, Spain) and used as received. Compositional details and physical properties
153 can be found elsewhere (Quinchia et al., 2010). Other general reagents used for
154 chemical analysis were of analytical grade and also acquired from Merck–Sigma-
155 Aldrich (Darmstadt, Germany).

156 *2.2. Production of electrospun SCG/PETbot nanofiber mats*

157 PETbot bottle pieces and SCG were dissolved in a 80:20 (v/v) TFA/DCM binary
158 solvent at room temperature (22 ± 1 °C) under magnetic stirring at 650 rpm for 24 h.
159 The total dissolved concentration was 15% (w/v) and the SCG/PETbot weight
160 proportion 1:0, 2:1, 1:1, 1:2 or 0:1. The sample code used for each proportion is shown
161 in Table S1 (see Supporting Information).

162 SCG/PTEbot solutions were used as feed streams in the electrospinning chamber
163 (DOXA Microfluidics, Málaga, Spain), which were continuously pumped at a
164 controlled flow rate of 0.8 mL/h through a plastic syringe furnished with a 20-G needle.
165 The syringe was horizontally arranged and connected to a high-voltage power source
166 that supplied 15 kV. Electrospun nanofibers were collected on an aluminum surface
167 positioned 10 cm from the needle tip. Electrospinning was conducted at 22 ± 1 °C and
168 $45 \pm 1\%$ relative humidity.

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170 *2.3. Preparation of oleogels*

171 Electrospun SCG/PETbot nanofiber mats were carefully removed from the collector and
172 directly dispersed at a 5 wt.% concentration in castor oil under gentle agitation (60 rpm)
173 at room temperature for 24 h before storage, also at room temperature. The oil was first
174 placed in an open vessel and nanofiber mats were slowly added and dispersed using an
175 Ika-RW 20 (Staufen, Germany) mixing device fitted with an anchor impeller.

176

177 *2.4. Characterization techniques*

178 The chemical composition of SCG was established after a hydrolysis treatment with
179 sulfuric acid at 30 °C. 5 mL of 72 wt.% sulfuric acid was added to an SCG aliquot (0.5
180 g) by gently stirring for 1 h. Subsequently, a post-hydrolysis with 4 wt.% sulfuric acid
181 (by adding water) was carried out in autoclave at 121 °C for 60 min. The resulting solid
182 residue was recovered by filtration and, based on Tappi T222, taken to be Klason lignin.
183 Monosaccharides in the hydrolyzates were determined by HPLC at 30 °C, with an
184 Aminex HPX-87H ion-exchange column as stationary phase and 0.005 M H₂SO₄ at a
185 flow rate of 0.6 mL min⁻¹ as mobile phase, according to Tappi T249 in order to estimate
186 the glucan, xylan and arabinan contents. An Eltra CHS-580A Helios device (Verder
187 Scientific, USA) was used for elemental analysis of SCG.

188 Fourier-transform infrared (FTIR) spectra for SCG, PETbot and nanofiber mats
189 were acquired with an FT/IR 4200 instrument from Jasco, Inc. (Tokyo, Japan) over the
190 wavenumber range 4000–400 cm⁻¹ (4 cm⁻¹ resolution and 200 scans per spectrum). For
191 this, the samples were placed in an attenuated total reflectance (ATR) accessory
192 furnished with a monolithic diamond crystal.

193 Thermogravimetric analyses (TGA) were carried out on a Q50
194 thermogravimetric analyzer from TA Instruments (New Castle, DE, USA). An amount
195 of 4–7 mg of sample was placed on a platinum pan and heated from 30 to 600 °C at 10
196 °C·min⁻¹ under an N₂ stream (flow rate, 100 mL/h).

197 The morphology of electrospun nanofiber mats was examined in a JXA-8200
198 SuperProbe scanning electron microscope (SEM) from JEOL (Tokyo, Japan), using an
199 acceleration voltage of 15 kV. The morphological observations were displayed at

200 $\times 4000$ and $\times 10\ 000$ magnifications. Previously, samples were gold sputtered in a
201 BT150 sputter coater from HHV (Crawley, West Sussex, UK). Oleogels were analyzed
202 microstructurally by using an AURIGA SEM apparatus from Zeiss (White Plains, NY,
203 USA), using an acceleration voltage of 20 kV. The oleogels were previously subjected
204 to chemical fixation (Pathan et al., 2010) and then sputtered with a thin layer of gold to
205 improve the quality of the observations (Stokroos et al., 1998). SEM images were
206 analyzed with the FIJI ImageJ software to determine some morphological parameters
207 such as the average fiber diameter. Each sample was analyzed from 100 random
208 observations at an identical magnification.

209 Electrospun nanofiber mats were subjected to uniaxial tensile tests on an AG-IS
210 Universal Testing Machine from Shimadzu Corp. (Kyoto, Japan), equipped with a 50 N
211 load cell, at room temperature (22 ± 1 °C). The extensional rate applied was $0.1\ \text{mm s}^{-1}$.
212 Samples were cut into rectangles 12.5×30 mm (~ 0.5 mm width) in size and placed in
213 the metal clamps with sandpaper to prevent slippage. The measurements thus made
214 allowed the maximum stress, strain at break and Young's modulus to be determined.

215 The rheological properties of the oleogels were examined with a Rheoscope
216 controlled-stress rheometer from Thermo Fisher Scientific (Waltham, MA, USA), using
217 a serrated plate-plate geometry (20 mm diameter, 1 mm gap). Viscous flow
218 measurements were conducted by applying an increasing stepped shear rate ramp (3 min
219 per point) from 10^{-2} – $10^2\ \text{s}^{-1}$. Small-amplitude oscillatory shear (SAOS) tests were
220 carried out within the linear viscoelastic region in the 0.03 – $100\ \text{rad s}^{-1}$ frequency range.
221 Prior to this, the extension of the linear viscoelastic range was determined by
222 performing stress sweep tests. All rheological measurements were made at 25 °C.

223 Tribological tests were conducted on a Physica MCR-501 rheometer from Anton
224 Paar (Graz, Austria) furnished with a tribological cell containing a steel ball 6.35 mm in
225 diameter that was rotated on three 45° inclined rectangular steel plates where the
226 oleogel samples were placed and tested as lubricants. A constant normal load of 20 N
227 on the inclined plates and a rotational speed of 10 rpm were applied for 10 min.
228 Tribological tests were performed under mixed lubrication and pure sliding conditions.

229 The tribological cell adapted to the rheometer allows the coefficient of friction to be
230 calculated as the ratio between the measured friction force and the applied normal load.
231 The effective normal force and sliding velocity were calculated from the applied axial
232 force, ball radius, and angular velocity as shown elsewhere (Heyer and Lauger, 2009).
233 The wear marks produced on the steel plates were examined with a BX51 Olympus
234 optical microscope (Tokyo, Japan) and the mean diameters were measured from the
235 images.

236

237 *2.5. Statistical analysis*

238 The data underwent ANOVA analysis, with a minimum of three replicates for each
239 independent measurement. Furthermore, a post hoc means comparison test was
240 conducted to reveal any statistically significant differences ($p < 0.05$).

241

242 **3. Results and discussions**

243 *3.1. Mat morphology*

244 Figure 1 shows the morphologies of selected nanostructures obtained by electrospinning
245 SCG/PETbot solutions in variable SGC:PETbot weight ratios. As can be seen,
246 electrospinning an PETbot-free SCG solution produced no nanofibers but only
247 microsized electrospayed particles (see Figs 1a and 1b). These results are consistent
248 with others previously obtained by electrospinning lignin fractions (Borrego et al.,
249 2021; Dallmeyer et al., 2010), which found that adding a cospinning or dopant polymer
250 was essential for uniform bead-free nanofibers to form.

251 Linear polymers such as PET are easier to electrospin than heterogeneous,
252 branched or low-molecular-weight polymers such as dissolved lignocellulosic fractions,
253 which cannot entangle properly (Rubio-Valle et al., 2021; Svinterikos et al., 2020).
254 Consequently, increasing the PETbot fraction in the electrospinning solutions helped
255 obtain and stabilize jet formation, thus significantly reducing the number of isolated or
256 embedded submicron particles. As shown by Figs 1c and 1d, adding PETbot to SCG in
257 a 1:2 SCG:PETbot ratio led to a highly homogeneous nanofiber network containing

258 some beaded nanofibers or embedded micrometric particles. Further increasing the
259 PETbot content (to a 1:1 SCG:PETbot ratio) almost completely avoided the formation
260 of beaded fibers and led to rather homogeneous nanofiber mats with an increased fiber
261 density (see Figs 1e and 1f) which, however, exhibited some surface texture or
262 roughness as a result of the Taylor cone still being unstable to some extent owing to the
263 contribution of SCG. This result is also consistent with those of other studies on the
264 electrospinnability of lignocellulosic materials (Baker and Rials, 2013; Dallmeyer et al.,
265 2014). Finally, the solution containing SGC and PETbot in a 1:2 weight ratio produced
266 highly uniform nanofiber mats (Figs 1g and 1h) that were very similar to those obtained
267 with PETbot alone (i.e., from the SCG-free solution) (Figs 1i and 1j), although they still
268 contained some nanofiber beads (see, for instance, Fig. 1h). Figure 1 also shows the
269 average nanofiber diameter as calculated from the SEM images. As can be seen, the
270 higher the PETbot proportion was, the higher was fiber diameter as a result of fewer
271 beads forming.

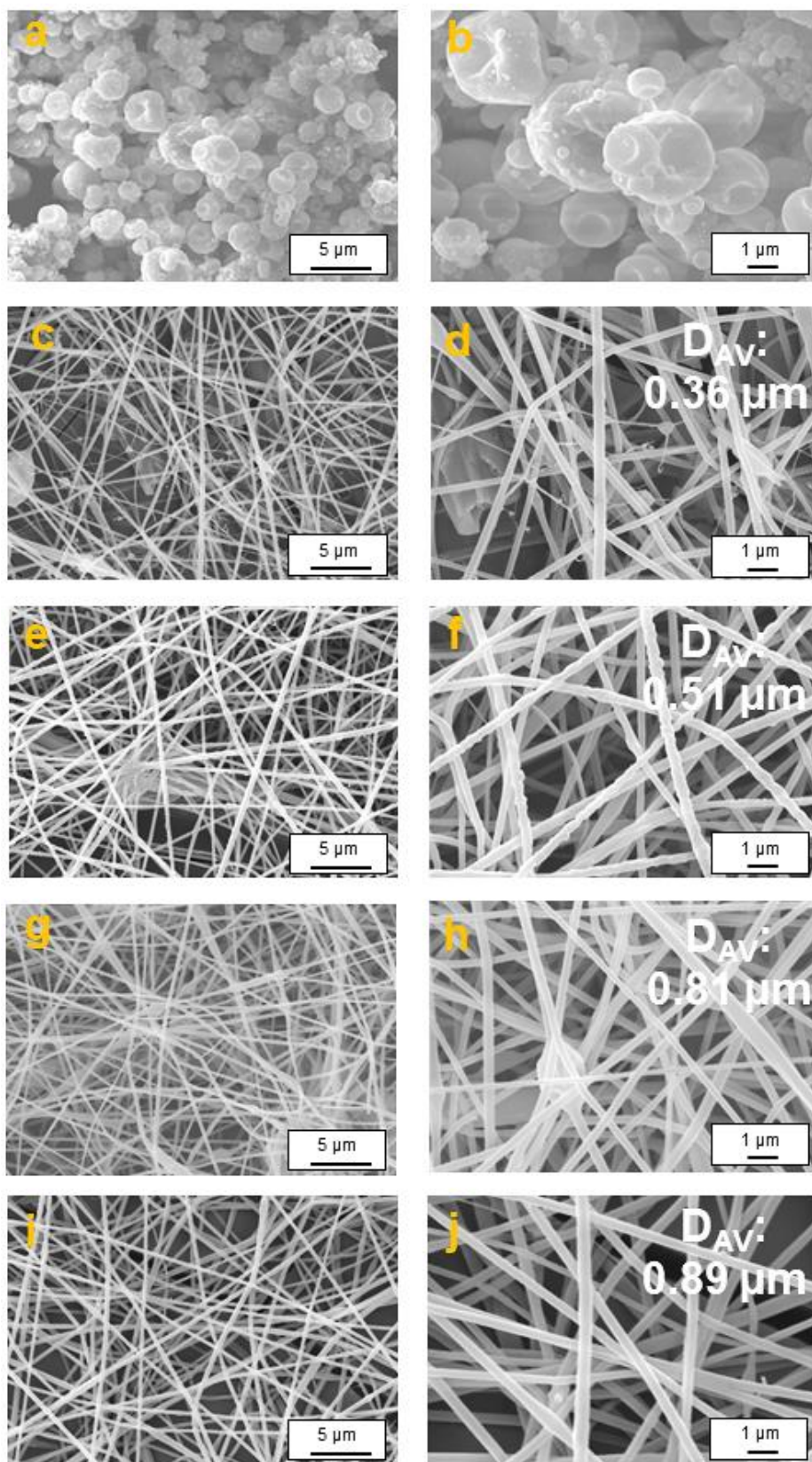


Figure 1. SEM images of SGCs/PETbot electrospun nanostructures obtained from solutions containing a 15 wt.% total polymer concentration in different SGC:PETbot weight ratios. (a), (b) 1:0. (c), (d) 2:1. (e), (f) 1:1. (g), (h) 1:2. (i), (j) 0:1. Magnification: $\times 4000$ (left), $\times 10\,000$ (right).

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273 3.2. Chemical and thermogravimetric analyses of the mats

274 The composition of the composite nanofiber mats was checked by FTIR spectroscopy
275 and TGA analysis. Figure S1 (Supporting Information) shows the FTIR spectra for
276 electrospun nanofiber mats obtained at different SGC:PETbot weight ratios as
277 compared with the starting materials (SCG and PETbot). The main peaks associated
278 with the chemical structure of PETbot were those corresponding to the ester C=O bond
279 at 1714 cm^{-1} ; the C=C stretching vibrations of the aromatic skeleton at 1577 and 1506
280 cm^{-1} ; those of the terephthalate group (OOC C_6H_4 -COO) at 1238 cm^{-1} , the methylene
281 group at 1090 cm^{-1} and the ester C-O bond at 1016 cm^{-1} ; and the wagging vibrations of
282 C-H bonds in the aromatic structure at 722 cm^{-1} (Pereira et al., 2017). Consistent with
283 its chemical composition (see Table S2, Supporting information), the FTIR spectrum for
284 SGC exhibited a broad band at 3355 cm^{-1} typically attributed to stretching vibrations in
285 O-H groups forming inter- and intramolecular hydrogen bonds in polymers such as
286 cellulose, hemicellulose and lignin; two bands at 2923 and 2849 cm^{-1} due to
287 asymmetric and symmetric C-H stretching vibrations; another two at 1515 and 1459
288 cm^{-1} due to aromatic skeleton vibrations in lignin; and a further two at 1070 and 1006
289 cm^{-1} that can be assigned to C-O bonds in lignin and cellulose, respectively (Davila-
290 Guzman et al., 2016). The presence of hemicellulose polysaccharides was confirmed by
291 a peak at 1737 cm^{-1} due to stretching vibrations in C=O groups (Nieva et al., 2019).

292 Both waste materials contributed to the FTIR spectra for the electrospun
293 SCG/PETbot nanofibers, which exhibited the above-described bands and were
294 consistent with those for sisal lignocellulose/PETbot fibers (Santos et al., 2015). For
295 instance, the broad band at ca. 3600 cm^{-1} corresponding to hydroxyl groups in SCG,
296 and those for C-H asymmetric and symmetric stretching vibrations at ca. 2918 and
297 2852 cm^{-1} , respectively, were quite apparent. Also, the peaks at ca. 1515 cm^{-1} and 1460
298 cm^{-1} can be assigned to aromatic skeletal vibrations in lignin and PETbot, while the
299 band at ca. 1721 cm^{-1} , corresponding to the C=O bond of hemicellulose acetyl groups,
300 was overlapped with that due to the ester group in PETbot. Bands due to the

301 terephthalate group and C–H vibrations in the aromatic structure of PETbot were also
302 apparent. More important, no signs of degradation of the starting waste materials were
303 detected.

304 Figure S2 (see Supporting Information) shows the TGA curves for electrospun
305 SCG/PETbot nanofibers and also those for the starting PETbot and SCG materials. The
306 temperature at the beginning of the degradation step (T_{onset}), that at which the
307 decomposition rate peaked (T_{max}) and the weight loss in each degradation step, as well
308 as the final proportion of residue remaining, were estimated from the thermograms for
309 the samples (see Table 1). Thermal degradation of PETbot from post-consumer plastic
310 bottles occurs in a single stage at $T_{\text{max}} = 401$ °C and results in a weight loss of 80%
311 corresponding to decomposition of the polymer (Das and Tiwari, 2019). On the other
312 hand, SCG degrades in three stages at a T_{max} of around 300, 333 and 395 °C, with a
313 mass loss of 31, 18 and 19%, respectively, through depolymerization and
314 decomposition of the main components of SCG (hemicellulose, cellulose and lignin;
315 Brachi et al., 2021). The profiles for the thermal degradation of electrospun
316 SCG/PETbot nanofibers, which also occurs in three stages if the initial weight loss due
317 to residual solvent is neglected, confirmed that the proportions of SCG and PETbot
318 were as intended. Thus, given that the main PETbot thermal degradation event roughly
319 overlapped with the third degradation stage of SCG over the T_{max} range 395–427 °C, an
320 increase in weight loss, together with a slight shift of the peak to 427 °C, was observed
321 as the PETbot proportion was raised. On the other hand, the weight loss in the second
322 thermal event (ca. 300 °C) increased with increasing SCG content. Finally, the
323 proportions of residue in the electrospun nanofibers were similar to those of PETbot but
324 lower than those of SCG, which indicates that part of the residue, especially in SCG,
325 was not dissolved in the electrospinning solution.

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Table 1. TGA characteristic parameters for electrospun SCG/PETbot mats and the starting waste materials (PETbot and SCG).

Samples	T_{onset} (°C)	T_{max} (°C)	Weight loss (%)	Residue (%)
PETbot	401	427	80	17
SCG	266/323/392	300/333/395	31/18/19	27
2SCG:1PETbot	158/271/415	240/300/399	8/30/40	17
1SCG:1PETbot	199/279/388	242/298/415	10/18/55	13
1SCG:2PETbot	208/282/398	248/300/425	7/10/65	14

331

332 3.3. Mechanical properties of the mats

333 Figure 2a illustrates the mechanical response of the nanofiber mats in the tensile tests.

334 All specimens exhibited a similar profile, with an initial linear region in the stress vs

335 elongational strain plot (i.e., an essentially elastic response), followed by an extensive

336 plastic region with a continuously decreasing slope. Once the maximum strain at failure

337 was reached, the specimens broke with a sudden decrease in stress. Figure 2b shows the

338 Young's modulus, maximum stress and tensile strain at break as calculated from the

339 plot. As can be seen, the three mechanical parameters increased with decreasing

340 SCG:PETbot weight ratio in the nanofibers. The Young's modulus of nanofiber mats

341 increased from 5.1 to 32.8 MPa as SGC:PETbot weight ratio decreased from 2:1 to 0:1,

342 whereas the strain at break and maximum stress values ranged from 13.8 to 39.4% and

343 from 6.6 to 33.1 MPa, respectively. The enhanced mechanical properties of nanofiber

344 mats as PETbot content increased was to be expected for two reasons, namely: (a) the

345 nature of PET, which is a linear thermoplastic polymer with excellent mechanical

346 performance (Owen et al., 2023) as compared with lignocellulosic polymers (Akato et

347 al., 2019; Xu et al., 2020); and (b) the architectural properties of the electrospun

348 nanofibers (Doderio et al., 2020), which, as stated above, were increasingly uniform,

349 bead-free and thicker as the PETbot proportion was increased. In particular, the average

350 fiber diameter strongly influences the extensional properties of the mats. Thus, both

351 Young's modulus and strain at break linearly correlated with fiber diameter, as shown

352 in Figs. 2c and 2d.

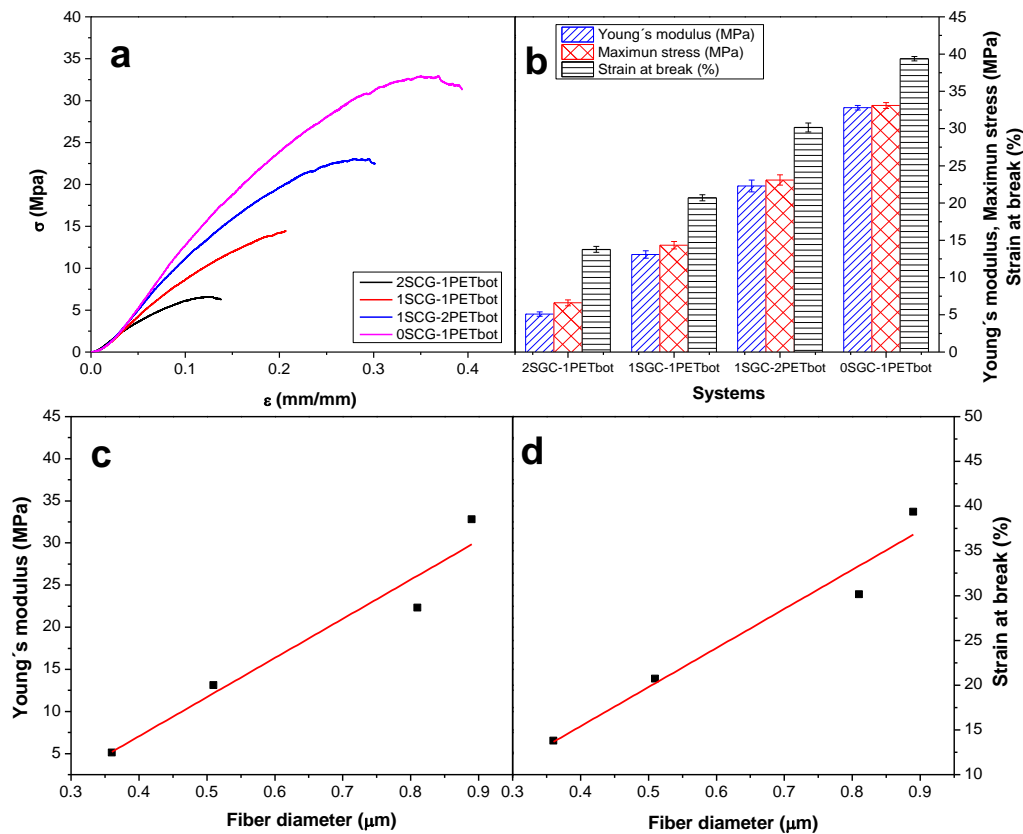


Figure 2. (a) Tensile stress vs elongational strain plots and (b) Young's modulus, maximum stress and tensile strain at break, of SGCs/PETbot electrospun nanostructures differing in SGC:PETbot weight ratio. Linear correlations ($R^2 > 0.932$) of Young's modulus (c) and strain at break (d) with average nanofiber diameter.

353

354 3.4. Oil structuring capability of electrospun SGC/PETbot nanofibers

355 Recent studies have shown that lignin nanostructures consisting largely of
 356 electrospun particles (i.e., nonfibrous nanostructures) give physically unstable
 357 dispersions in vegetable oils (Borrego et al., 2022; Rubio-Valle et al., 2023, 2021),
 358 whereas electrospun (bead-free or even slightly beaded) nanofibers can generate
 359 physically stable oleogels by forming percolation networks through enhanced physical
 360 interactions arising from the high specific surface area and aspect ratio of the
 361 nanofibers. The electrospun SGC/PETbot nanofiber mats obtained from the waste
 362 materials used here dispersed easily in castor oil to form gel-like dispersions visually
 363 resembling those of lubricating greases or oil based coating precursors (Borrero-López
 364 et al., 2021; Gallego et al., 2015b). Figure 3 shows the SEM morphology of the oleogel

365 microstructures obtained by dispersing 5 wt.% electrospun SCG/PETbot nanofibers in
366 castor oil. As can be observed, the oleogels obtained with nanofiber webs consisting of
367 2:1 SGC:PETbot (w/w) exhibited a more heterogeneous microstructure and less
368 uniform distribution of fiber diameters than did those obtained with lower SGC:PETbot
369 weight ratios. Upon dispersion in castor oil, the nanofibers agglomerated and swelled to
370 an increasing extent with increasing SGC:PETbot weight ratio. Table 2 shows the
371 degree of nanofiber swelling as calculated from the average diameters obtained from
372 SEM images of electrospun nanofiber networks (Fig. 1) and oleogel microstructures
373 (Fig. 3), and defined as (Rubio-Valle et al., 2023):

$$374 \quad \beta = \frac{D_f - D_o}{D_o} \times 100 \quad (1)$$

375 where D_o denotes the average diameter of electrospun nanofibers and D_f that of the
376 fibers upon dispersion in the oil medium. As can be seen, the degree of swelling (i.e.,
377 the fiber diameter of the percolation network) increased markedly with increasing
378 SGC:PETbot weight ratio (up to 92% with 2:1), but remained essentially unchanged in
379 SCG-free PETbot nanofibers. Therefore, diffusion of the vegetable oil into nanofibers
380 and the associated swelling occurred mostly in SCG biopolymers (Onwuka et al., 2018,
381 2016). Since castor oil is relatively polar, these results can be ascribed to the
382 hydrophilicity of SCG biopolymers facilitating penetration of triglycerides present in
383 the oil into the fibers and subsequent H-bonding (Boey et al., 2022). This cannot have
384 been the case with PETbot owing to its hydrophobicity (Jasmee et al., 2018; Shukla et
385 al., 2009).

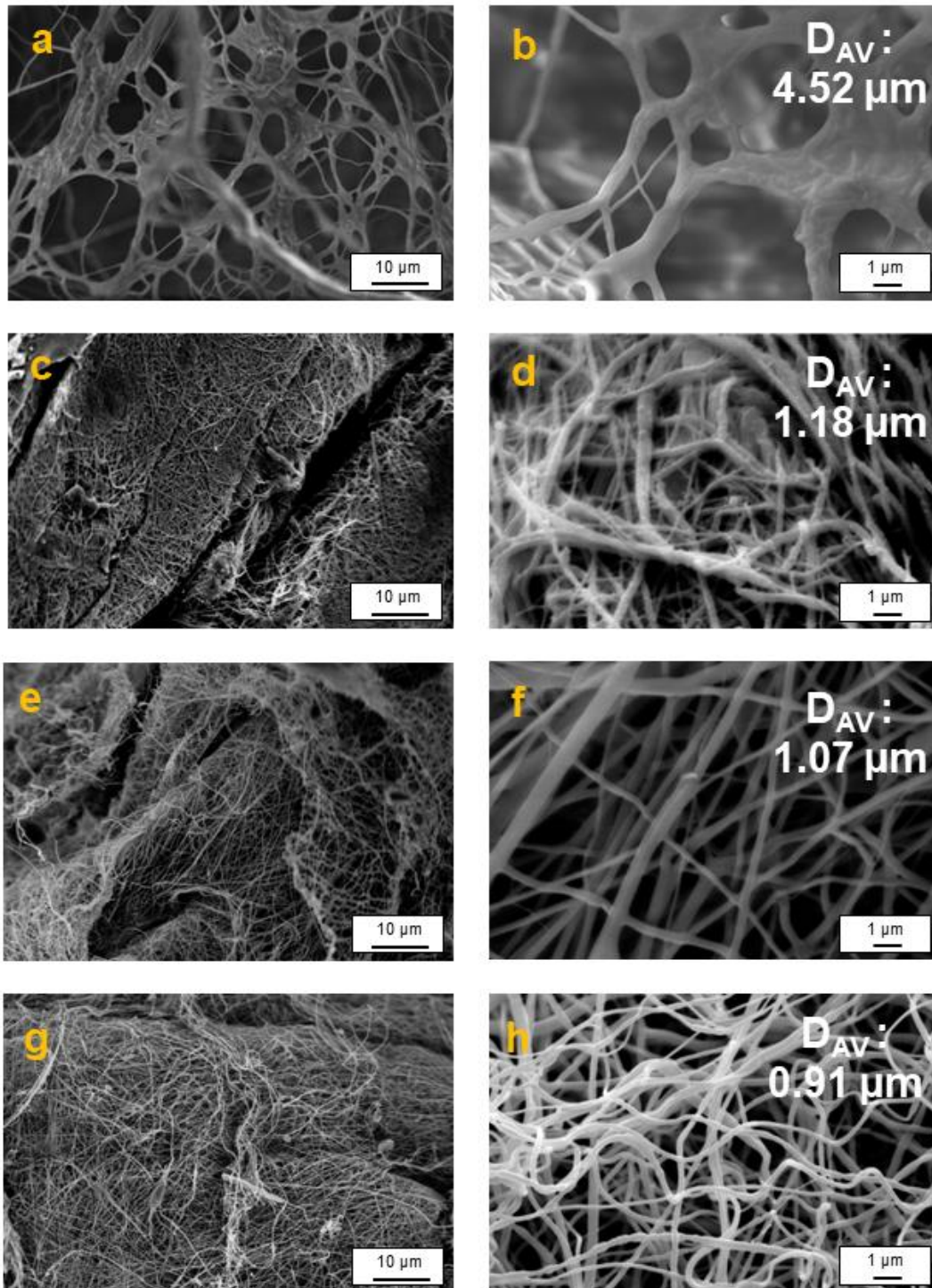


Figure 3. SEM morphological observations of oleogels prepared with electrospun SGC:PETbot nanostructures as a function of the SGC:PETbot weight ratio. (a), (b) O-2SGC-1PETbot. (c), (d) O-1SGC-1 PETbot. (e), (f) O-1SGC-2PETbot. (g), (h) O-0SGCs-1PETbot. Magnification: $\times 1000$ magnification (left), $\times 10\,000$ (right).

386

387

Table 2. Swelling factor (eq. 1), and consistency and flow index, for oleogels prepared with electrospun SCG/ PETbot nanostructures, as well as friction coefficient values and wear scar diameters obtained by using the oleogels as lubricants in a ball-on-plates tribological contact

Oleogel	Swelling factor β (%)	K (Pa·s ⁿ)	n (-)	Friction coefficient (-)	Wear scar diameter (μm)
O-2SGC-1PETbot	92.03 ^a	45.57 ^A	0.14 ^{aA}	0.165 ^{α}	421 ^{$\alpha\alpha$}
O-1SGC-1PETbot	56.78 ^b	98.31 ^B	0.09 ^{bB}	0.145 ^{β}	365 ^{$\beta\beta$}
O-1SGC-2PETbot	24.29 ^c	190.89 ^C	0.13 ^{aA}	0.117 ^{γ}	335 ^{$\gamma\gamma$}
O-0SGC-1PETbot	2.19 ^d	717.36 ^D	0.06 ^{bB}	0.092 ^{γ}	315 ^{$\gamma\gamma$}
Commercial NLGI 2 lithium grease *	-	-	-	0.113	273
Commercial NLGI 2 calcium grease *	-	-	-	0.112	321

Values on the same column differing in their superscripts were significantly different ($p < 0.05$)

* data taken from Gallego et al. (2016)

388

389 Figures 4a–4c show the variation of the SAOS functions (viz., the storage modulus, G' ,
390 the loss modulus, G'' , and the loss tangent, G''/G') with frequency in the oleogels
391 obtained by dispersing 5 wt% electrospun SCG/PETbot nanofiber mats in castor oil at
392 variable SCG:PETbot ratios. As can be seen, all samples gave very similar mechanical
393 spectra typical of gel-like dispersions that differed only in the magnitude of G' and G''
394 —which increased by almost two decades with decreasing SCG:PETbot ratio. On the
395 other hand, the loss tangent, indicative of the relative elastic character, differed little but
396 was slightly lower in nanofibers consisting mainly of PETbot (1:2 and 0:1 SCG:PETbot
397 ratios), especially at high frequencies (see Fig. 4c). This viscoelastic behavior
398 corresponds to the so-called plateau zone of the SAOS functions vs. frequency plots,
399 where G' exceeds G'' over an extended frequency range and is almost independent of
400 the frequency whereas G'' tends to a minimum. An extended plateau region of the
401 mechanical spectrum is typical of physical entanglement in polymeric materials (Ferry,
402 1980) and, in our case, reflects the packing effect of the percolation network formed by

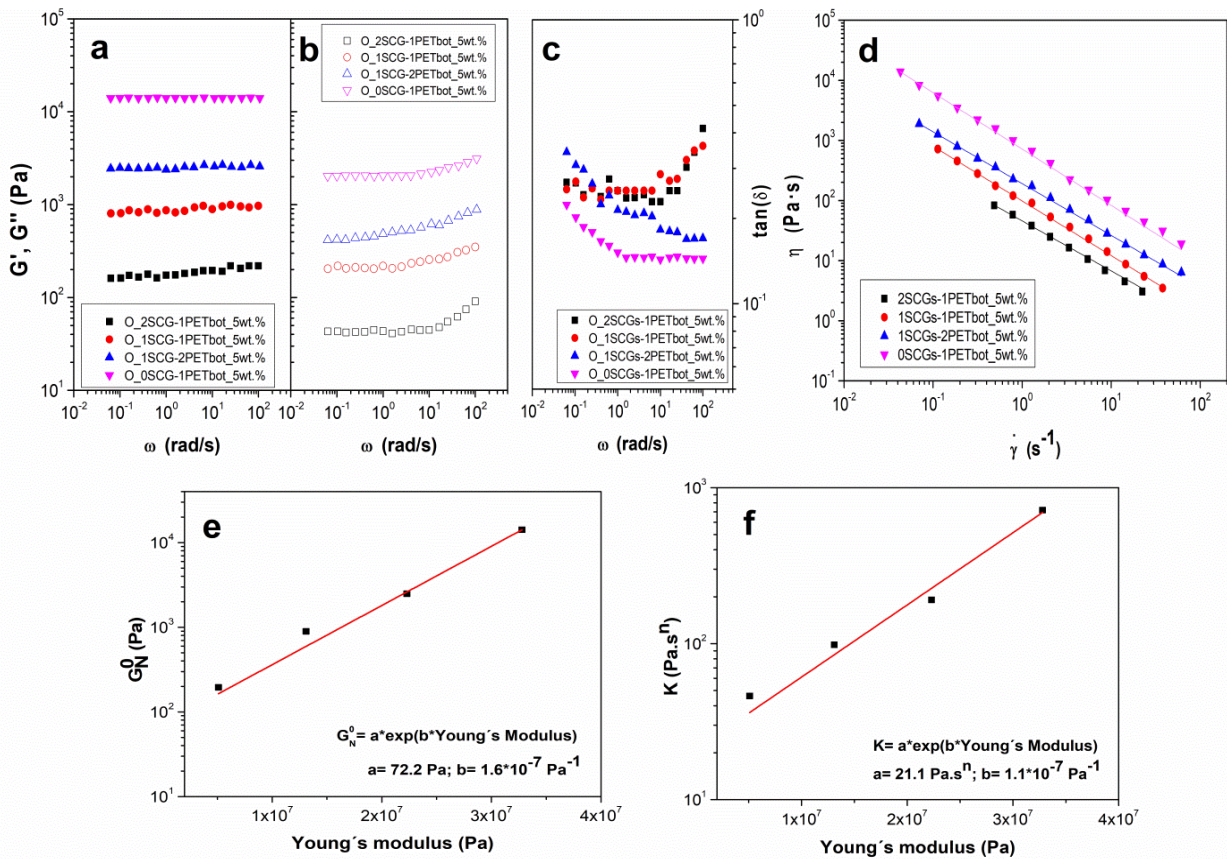
403 electrospun fibers. A very similar rheological response is exhibited by conventional
404 lubricating greases, with typical G' values of 10^3 – 10^5 Pa depending on the thickener
405 type and concentration, and G'' values roughly one order of magnitude lower (Sánchez
406 et al., 2014). This mechanical spectrum is characterized in rheological terms by the
407 plateau modulus (G_N^0), which is a measure of entanglement density and hence of gel
408 strength (Baurngaertel et al., 1992). The observed increase in gel strength with
409 decreasing SCG:PETbot ratio can be ascribed to the intrinsic mechanical properties of
410 the nanofiber mats (specifically, to their Young's modulus). As can be seen in Fig. 4e,
411 the plateau modulus (G_N^0) of the oleogels increased exponentially —linearly on a log
412 scale— with increasing Young's modulus in the electrospun nanofiber mats, which in
413 turn correlated with the average nanofiber diameter (see Fig.2c).

414 Regarding the viscous flow response, Fig. 4d shows the viscosity vs. shear rate
415 plots for oleogels prepared with electrospun nanofibers differing in SCG:PETbot ratio.
416 A shear-thinning behavior was invariably observed over the experimental shear rate
417 range studied that fitted the power-law model:

$$418 \qquad \eta = K \cdot \dot{\gamma}^{n-1} \qquad (2)$$

419 where K and n are the consistency and flow indices, respectively. Table 2 shows the K
420 and n values obtained by fitting the data to eq. (2). Similarly to SAOS functions, the
421 lower was the SCG:PETbot weight ratio of the nanofiber mats, the higher was K .
422 However, the low values of the flow index, n , reveal strong shear-thinning responses in
423 all cases. As previously shown for G_N^0 , K was also exponentially correlated with
424 Young's modulus in the nanofiber mats. Overall, the morphology and mechanical
425 properties of the electrospun nanoarchitectures governed the rheological properties of
426 the oleogels and can be tuned through the SCG:PETbot ratio. It should be noted that the
427 rheological response of oleogels containing only 5 wt.% nanofibers was comparable to
428 that of conventional lubricating greases with much higher thickener contents (e.g.,
429 lithium soaps; Delgado et al., 2006).

430



4.
 432 **Figure 4.** Influence of the SCG:PETbot weight ratio and the mechanical properties of the
 433 associated nanostructure on the rheological properties of the oleogels. Variation of
 434 (a) storage modulus (G'), (b) loss modulus (G'') and (c) loss tangent with frequency; (d)
 435 variation of viscosity with shear rate, and correlations ($R^2 > 0.981$) of (e) the plateau
 436 modulus and (f) the consistency index of the oleogels with Young's modulus for the
 437 nanofiber mats.

439 3.5. Lubrication performance of the oleogels

440 Bearing in mind the potential use as lubricants of the oleogels, their lubrication
 441 performance was assessed by measuring friction and wear scar in a tribological
 442 steel–steel contact. For this purpose, friction coefficients were recorded at constant
 443 normal load (20 N) and rotational speed (10 rpm) over time. Their average values are
 444 shown in Table 2 together with the average diameter of the wear marks generated on the
 445 plates upon completion of the friction test. As can be seen, both friction and wear
 446 dimensions were significantly reduced by mats with increased PETbot contents. As
 447 stated above, this was the likely result of nanofibers with increased PETbot ratios being
 448 much less prone to swelling in the oleogel, thus allowing the oil to be released more
 449 easily into the lubricated contact. In addition, the higher viscosity of oleogels prepared

450 with nanofibers of lower SCG:PETbot ratios (see Fig. 4d) must favour the increase of
451 the lubricant film thickness thus preventing friction and wear.

452 On the other hand, most of the oleogels exhibited acceptable friction coefficients and
453 antiwear performance when used as lubricants, comparable to those found with
454 commercial lubricating greases (Gallego et al., 2016; Zhang et al., 2018) and other
455 oleogels produced with chemically modified biopolymers (Cortés-Triviño et al., 2019)
456 under similar conditions. For the sake of comparison, Table 2 also shows friction
457 coefficient and wear diameter data (Gallego et al., 2016) obtained when using
458 conventional NLGI 2 lithium and calcium greases as lubricants, tested in the same
459 tribological cell under identical conditions as the oleogels. As can be observed, the
460 friction coefficient values measured with oleogels prepared with nanofiber mats of low
461 SCG:PETbot ratios (1:2 and 0:1) were similar to or even lower than those found with
462 these commercial greases. Regarding wear marks, most of the SCG/PETbot nanofibers-
463 based oleogels provided a wear prevention functionality comparable to that shown by
464 the calcium grease.

465

466 *3.6. Outlook*

467 *3.6.1. Limitations and further work*

468 This work demonstrates that the methodology used to obtain nanofibers with the ability
469 to structure oil from waste materials such as SCG and PETbot is effective and relatively
470 simple, and avoids any chemical modification of the structuring agent to enhance the
471 compatibility with oil. However, this methodology is not without limitations and future
472 work should be directed towards overcoming these. One of the main drawbacks of
473 electrospinning to be implemented at a large scale is its low production efficiency due to
474 the usually low feeding rate and possible nozzle clogging. Certainly, nowadays there are
475 a number of alternatives available based on multiple needles in parallel and needle-less
476 arrangements that allow an affordable scale-up of this technology (Hernandez et al.,
477 2022; Shi and Giapis, 2018; Xu et al., 2023). Nevertheless, these configurations can
478 improve efficiency to a certain extent but still cause some other problems such as multi-

479 needle electric field interference, generation of inhomogeneous fibers, and/or poor fiber
480 distribution in the collector (Xue et al., 2019), while for needle-less electrospinning the
481 stability of the electrospun jet and controllability of the equipment are far lower than
482 that of needle-based electrospinning equipments (Si et al., 2023). Therefore, scale-up
483 tests need to be performed to assess the efficiency and effectiveness of this technology
484 to produce SCG/PETbot nanofibers at a large scale. On the other hand, the proposed
485 methodology consists of a two-step oleogel manufacturing process, firstly obtaining the
486 nanofiber mats and, subsequently, the efficient dispersion in the oily medium. From a
487 cost-efficiency point of view, the integration of the two steps in a single operation is a
488 challenge that should be addressed, for example by designing a collector system in the
489 electrospinning chamber that includes the direct incorporation of the nanofibers into the
490 oil.

491 The other main limitation of this methodology is the use of harmful solvents
492 such as TFA to dissolve SCG and feed the electrospinning chamber. SCG as a complex
493 lignocellulosic waste material is difficult to dissolve. In this respect, there is still much
494 room for improvement in the search for environmentally friendly solvents suitable for
495 these hard-to-dissolve lignocellulosic materials that can also be easily electrospinnable.
496 In this sense, alternatives such as ionic liquids and NADES could be envisaged.

497 Finally, a more exhaustive tribological and thermo-rheological characterization
498 of the oleogels prepared with electrospun SCG/PETbot nanofibers should be performed
499 to ensure a proper implementation as lubricants.

500

501 *3.6.2. Contribution to the sustainable development*

502 Despite the aforementioned limitations, this work explores for the first time the use of
503 spent coffee grounds (SCG) and post-consumer bottles (PETbot) to obtain electrospun
504 nanofiber mats with a view to valorizing or upcycling these environmentally relevant
505 waste materials in high added-value applications which require oil structuring. As
506 mentioned in the Introduction section, the finding of suitable environmentally friendly,
507 renewable and/or recycled oil structuring and thickening agents is challenging for

508 several industrial sectors. For instance, the proposed approach could provide a new
509 niche market for some industrial sectors such as lubricant and coating production, where
510 a pressing need exists to replace conventional thickening agents for oils. Moreover, the
511 use of these kinds of waste materials as structuring agents is likely to yield more cost-
512 effective formulations than those based on metal soaps and polyureas.

513 Therefore, upcycling SCG and PETbot into nanofiber mats for oil structuring
514 applications is fully aligned with the UN Sustainable Development Goals (SDG), and
515 more specifically with the 12th SDG related to “Response Consumption and Production”
516 which clearly establishes targets such as substantially reducing waste generation
517 through prevention, reduction, recycling and reuse; achieving the environmentally
518 sound management of chemicals and all wastes throughout their life cycle; or
519 strengthening the scientific and technological capacity to move towards more
520 sustainable consumption and production patterns.

521

522 **4. Conclusions**

523 From the experimental results, the following conclusions can be derived:

- 524 • Electrospinning SCG solutions did not generate nanofibers but microsized
525 electrosprayed particles. However, adding PETbot as a cospinning polymer provided
526 homogeneous nanofiber networks with some beaded nanofibers. Increasing the
527 PETbot:SCG weight ratio above 1:1 prevented the formation of beads and increased
528 the average fiber diameter. As a result, the tensile properties of the nanofiber mats
529 improved with increasing proportion of PETbot in the nanofibers.
- 530 • Dispersions of electrospun SCG/PETbot nanofibers in castor oil exhibited gel-like
531 viscoelastic responses and shear-thinning characteristics comparable with those of
532 conventional lubricating greases, and also tribological properties suitable for their
533 use as lubricants.
- 534 • The rheological properties of the oleogels depend on the morphology and mechanical
535 properties of the electrospun SCG/PETbot nanofiber mats. Linear viscoelastic
536 functions, relative elasticity and shear viscosity increased with increasing proportion

537 of PETbot in the mats. The plateau modulus and the consistency index of the
538 oleogels correlated exponentially with Young's modulus in the mats.

539 • Overall, SCG and PETbot can be used to obtain electrospun nanofiber mats with oil
540 structuring capabilities. The resulting oleogels can be used as environmentally
541 friendly alternatives to conventional lubricating greases and their rheological
542 properties tuned through the SCG:PETbot weight ratio.

543

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550

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